Crystal Growth: Physics, Technology and Modeling

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Lecture 5. Thermodynamics of growth processes

http://w3.unipress.waw.pl/~stach/cg-2022-23/

Equilibrium

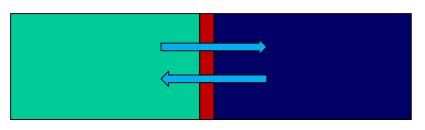
Equilibrium - extent

- Between different phases
- Inside single phase:
 - spatial local equilibrium
 - degrees of freedom partial equilibrium

Equilibrium - type

- mechanical
- thermal
- chemical

Phase equilibrium - fluxes



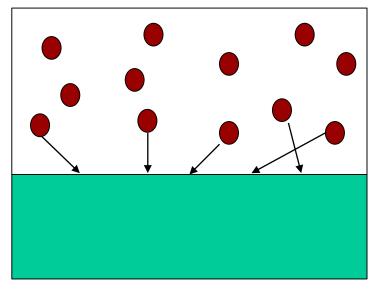
$$\overrightarrow{g} \equiv \frac{\Delta G \overrightarrow{n}}{\Delta S \Delta t}$$
 Flux

 \vec{n} = unit vector, normal to the surface

- Two phase state, stationary, i.e. time independent, such that introduction of barriers does not enforce any change
- Flux vectorial quantity \vec{g} indicating amount of physical quantity (conserved ΔG) transported across unit of surface (ΔS) in the unit of time (Δt)
- Two phase state, stationary, i.e. time independent: fluxes are balanced
 - mechanical → momentum flux
 - thermal \rightarrow energy flux
 - chemical \rightarrow mass flux

Fluxes - balance

Two phases – solid and vapor



• Molecule flux *I* (number of molecules arriving at the unit area of the surface in the unit of time

$$I = \frac{p}{\sqrt{2\pi mkT}}$$

Ideal gas – equilibrium and transport properties

Equation of state

$$pV = nkT$$
 $p = nkT$ $k = 1.3800648 \times 10^{-23} J/K$

• Mean free path λ and mean free flight time τ , thermal velocity $\langle v \rangle$

$$\lambda = \frac{1}{\sqrt{2}n\sigma} \qquad \tau = \frac{\lambda}{\langle v \rangle} \qquad \langle v \rangle = \sqrt{\frac{3kT}{m}}$$

Average molecular flux – does not depend on transport properties

$$I = \frac{p}{\sqrt{2\pi mkT}} = n \sqrt{\frac{kT}{2\pi m}}$$

Standard and normal conditions (IUPAC & NIST)

• Standard conditions (to 1982)

$$T = 273.15 K$$

$$p = 1$$
 atm = 101325 Pa

• Standard conditions (to 1982)

$$T = 273.15 K$$

$$p = 1 \text{ bar} = 100000 \text{ Pa}$$

Normal conditions

$$T = 293.15 K$$

$$p = 1$$
 atm = 101325 Pa

Ideal gas - numbers

• Density *n*

$$n = 2.687 \times 10^{19} \, cm^{-3}$$

• Average thermal velocity $\langle v \rangle$ at T = 300 K

$$\langle v \rangle = \sqrt{\frac{3kT}{m}} = 512 \ m/s$$

• Mean free path λ and mean free flight time τ

$$\lambda = \frac{1}{\sqrt{2} n \sigma} = 3 \times 10^{-6} m \qquad \tau = \frac{\lambda}{\langle v \rangle} = 5.6 \times 10^{-9} s$$

Molecule size - $d = 2R \sim 10^{-10} m$

Total scattering crossection- $\sigma = \pi R^2 \sim 10^{-20} m^2$

Surface geometry

GaN density (molecules) n_{GaN}

$$n_{GaN} = 4.3 \times 10^{22} \ cm^{-3}$$

GaN molar volume v_{GaN}

$$v_{GaN} = 2.3 \times 10^{-23} cm^3$$

• GaN surface atom density η_{GaN}

$$\eta_{GaN} = \eta_{GaN}^{2/3} = 8.671 \times 10^{14} \ cm^{-2}$$
 $\eta_{GaN} = \frac{4}{3a^2\sqrt{3}} = 7.611 \times 10^{14} \ cm^{-2}$

• GaN surface area for single site (0001) surface η_{GaN}

$$\varsigma_{GaN} = v_{GaN}^{2/3} = 1.145 \times 10^{-15} \ cm^2$$

$$\varsigma_{GaN} = \frac{3a^2\sqrt{3}}{4} = 1.314 \times 10^{-15} \ cm^2$$

GaN lattice constant – a = 3.1890 Å

Fluxes & surface equilibrium

Average molecular flux I

$$I = \frac{p}{\sqrt{2\pi mkT}} = 2.785 \times 10^{23} \, s^{-1} cm^{-2}$$

Site impact frequency v

$$v = I\varsigma_{GaN} = 3.788 \times 10^8 \, s^{-1}$$

Adsorption rate r

$$r = \nu \sigma = I \varsigma_{GaN} \sigma$$

 σ – sticking coefficient i.e. probability of attachment $(0 \le \sigma \le 1)$

Sticking coefficient

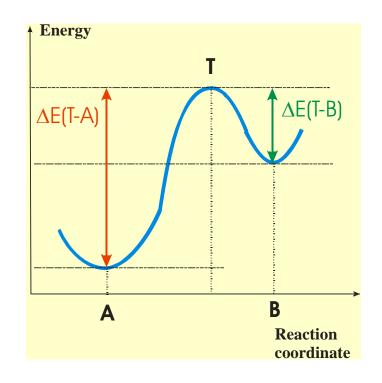
• Sticking coefficient – probability of transition P from initial (I - vapor) to final (F - surface attached) state

$$\sigma \equiv P (I \rightarrow F)$$

- Probability of transition P is calculated using notion of transition state T
- Transition state *T*
 - No return from Transition state (T) to Initial state (I)
 - Energy higher than initial state
 - Energy landscape saddle point

Transition state – time evolution of the system

• Time evolution chemical kinetics approach



- Energy barriers
- $-A \rightarrow B$ transition

$$\Delta E(T-A) = E(T) - E(E)$$

- B \rightarrow A transition

$$\Delta E(T-B) = E(T) - E(B)$$

Energy – total energy of the system in adiabatic approximation

Transition state –probability of arrival

Tolman principle

- Probability of the system in a given macro state is proportional to the volume of the phase space Q compatible with this macro state
- For reaction coordinate the relative probability is:

$$\frac{P(T)}{P(A)} = \frac{Q(T)}{Q(A)} exp \left[-\frac{F(T) - F(A)}{kT} \right]$$

Q(T) , Q(A) statistical sums over remaining degrees of freedom in states T and \boldsymbol{A}

Free energy of the system of the temperature T:

$$F = -kB \ln Q = -kT \ln(Q_{tr}Q_{viv}Q_{rot})$$

Transition state – Arrhenius principle

Absence of strong coupling (energy coupling on Hamiltonian level)

$$\frac{P(T)}{P(A)} = exp\left[-\frac{F(T) - F(A)}{kT}\right]$$

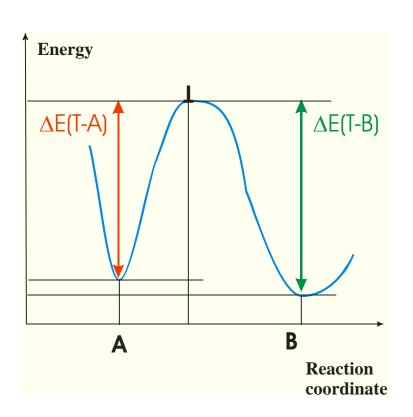
$$\frac{P(T)}{P(A)} = exp\left[\frac{S(T) - S(A)}{k}\right] exp\left[-\frac{E(T) - E(A)}{kT}\right]$$

No entropy difference:

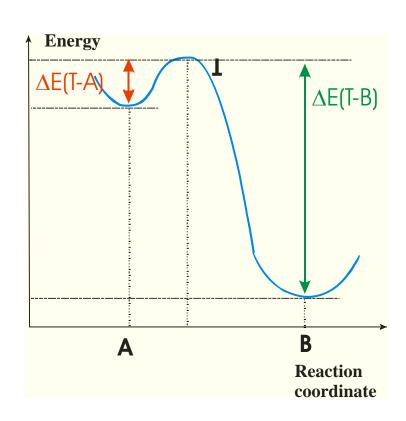
$$\frac{P(T)}{P(A)} \cong exp\left[-\frac{E(T) - E(A)}{kT}\right] = exp\left[-\frac{\Delta E}{kT}\right]$$

Energetic properties of the growth models

• Dynamic models



Kinetic models



$$r = \nu P(I \rightarrow F) = \nu exp\left[-\frac{\Delta E}{kT}\right]$$

Growth models

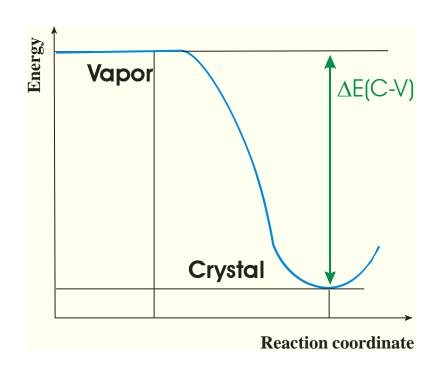
Dynamic models

- Transition forward and backward have comparable probability
- Both direction are important
- Existence of equilibrium state

Kinetic models

- o Probabilities are drastically different
- Alternatively blocking of some transition is enforced
- Absence of equilibrium state

Vapor – crystal growth models



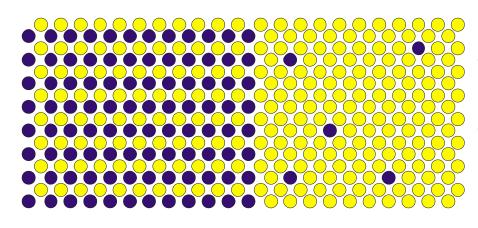
• Deposition at the single site $(v \rightarrow c)$

$$r = I \varsigma = \frac{p\varsigma}{\sqrt{2\pi mkT}} = n\varsigma \sqrt{\frac{kT}{2\pi m}}$$

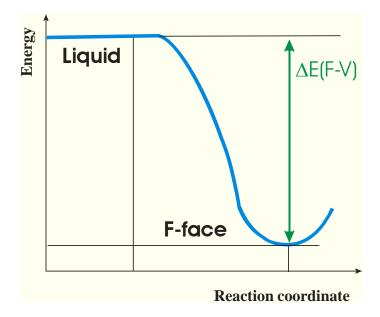
• Sublimation $(c \rightarrow v)$

$$r = \nu exp \left[-\frac{\Delta E(C-V)}{kT} \right]$$

Liquid – crystal growth models



- Diffusion
- Crystallization $(l \rightarrow c)$



• Dissolution $(c \rightarrow l)$

$$r = \nu exp \left[-\frac{\Delta E(C-L)}{kT} \right]$$

Growth dynamics – deviation from equilibrium

- **Equilibrium**
 - Dynamical

Thermal

Chemical

$$p_v = p_l = p_{eq}$$

$$T_v = T_l = T_{eq}$$
 $\mu_v = \mu_l = \mu_{eq}$

$$\mu_v = \mu_l = \mu_{eq}$$

- **Deviation from equilibrium**
 - o **Supersonic**

- SupercoolingSupersaturation

$$\Delta \tau = \frac{T_{l,v} - T_{eq}}{T_{eq}}$$

$$\Delta \mu = \frac{\mu_{l,v} - \mu_{eq}}{kT}$$

$$\Delta \mu = \frac{\mu_{l,v} - \mu_{eq}}{kT}$$

Supersaturation σ in ideal systems

Ideal gas

$$\sigma = \frac{\Delta \mu}{kT} = \ln \left(\frac{p}{p_{eq}} \right) = \ln \left(1 - \frac{p - p_{eq}}{p_{eq}} \right) \cong \frac{p - p_{eq}}{p_{eq}}$$

Ideal solution

$$\sigma = \frac{\Delta \mu}{kT} = \ln \left(\frac{x}{x_{eq}} \right) = \ln \left(1 - \frac{x - x_{eq}}{x_{eq}} \right) \cong \frac{x - x_{eq}}{x_{eq}}$$

Dynamic processes - metastable states – nucleation barriers

- Some systems remain in mother state under supersaturation energy barriers (height of the transition state)
- Transition probability of attainment of transition state
- Nucleation theory transition state is nucleus of the new phase (solid) of sufficiently large size
- Isotropic case finite size system is represented by the sphere of the radius R. The energy barrier (energy of the nucleus is):

$$\Delta E = \frac{4\pi R^3 \rho \ \Delta \mu}{3} + 4\pi R^2 \ \gamma$$

 $\Delta \mu = -\sigma < 0$ – volume term is negative

 $\gamma > 0$ – surface term is positive

Nucleation barriers - size

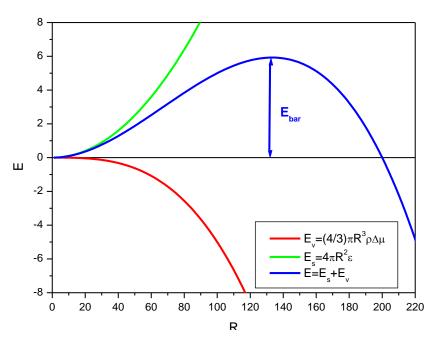
$$\Delta E = \frac{4\pi R^3 \rho \ \Delta \mu}{3} + 4\pi R^2 \ \gamma$$

Surface term dominates for small R

Volume term dominates for large R

Nucleation critical radius

$$R_{crit} = \frac{2 \gamma}{\rho |\Delta \mu|} = \frac{2 \gamma}{\rho k T \sigma}$$



Energy barrier - energy at critical radius

$$\Delta E = \frac{16\pi\gamma^3}{3\rho^2(\Delta\mu)^2} = \frac{16\pi\gamma^3}{3\rho^2(kT\sigma)^2}$$

Nucleation types

- Heterogeneous nucleation process in which the energy barrier is lowered due to existence of other factors (third phase, etc.)
- Homogenous nucleation process with standard energy barrier

Nucleation rate r_{nucl}

$$r_{nucl} = \nu N_s P(R_{crit})$$

Probability of critical nucleus $P(R_{crit})$

$$P(R_{crit}) = exp\left[-\frac{\Delta E_{crit}}{kT}\right] = exp\left[-\frac{16\pi\gamma^{3}}{3\rho^{2}(\Delta\mu)^{2}kT}\right]$$

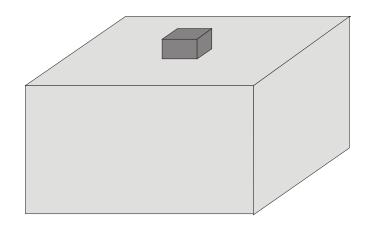
Number of sites at the surface N_s

$$N_s = \frac{4\pi R_{crit}^3 \rho_s}{3}$$

Attempt frequency **v**

Crystal growth rate – nucleation controlled

Crystals may be controlled by creation of new atomic layers – via
 2-d nucleation



$$r_{nucl} = v N_L P(R_{crit}) = v N_L exp\left(-\frac{\Delta E_{crit}}{kT}\right)$$

Number of sites at the Edge - N_L

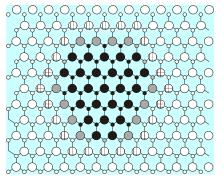
$$N_L = \frac{L}{a}$$

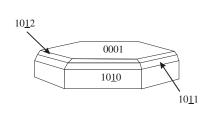
Edge energy for lattice constant – energy of broken bond - ϕ

Edge length - L

Lattice constant - a

2-d nucleation on GaN(0001) & GaN(10-10) surfaces





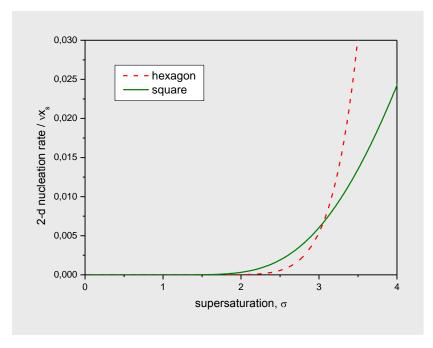
1011

1010

$$I = \nu \left(\frac{6\phi}{|\Delta\mu|} + 6 \right) exp \left[-\frac{1}{kT} \left(\frac{3\phi^2}{|\Delta\mu|} - \frac{3|\Delta\mu|}{4} \right) \right]$$



$$I = v \left(\frac{4\phi}{|\Delta\mu|} - 1 \right) exp \left[-\frac{1}{kT} \left(\frac{2\phi^2}{|\Delta\mu|} - \frac{|\Delta\mu|}{8} \right) \right]$$



- Growth anisotropy, i.e. different rates leading to different shapes
- Acceleration of the growth along [0001] direction for high supersaturation

Kinetic process – diffusion

Kinetic process – diffusion, i.e. random motion of the species (atoms or molecules):

- Mean (averaged over many jumps) translation is zero
- Correlation of the direction of the two consecutive jumps is zero (Markov process, i.e. process with no memory)

Diffusion types:

- Free jumps directions and lengths are purely random
- Lattice jumps between lattice sites

Free diffusion in the vapor

- Mean average translation is zero (no convection)
- Correlation of the direction of the two consecutive jumps is zero (Markov process, i.e. process with no memory)
- Mean free path
- Mean free flight time
- Average thermal velocity

$$\lambda = \frac{1}{\sqrt{2} n \sigma}$$

$$au = \frac{\lambda}{\langle v \rangle}$$

$$\langle v \rangle = \sqrt{\frac{3kT}{m}}$$

Total scattering crossection - σ

Lattice diffusion

- Species localized in lattice sites
- Correlation of the direction of the two consecutive jumps is zero (Markov process, i.e. process with no memory)
- The rate *r* is given by

$$r = \nu P(i \rightarrow f)$$

Attempt frequency **v**

• Jump probability

$$P(i \rightarrow f) = \frac{1}{C} exp \left[-\frac{\Delta E_{diff}}{kT} \right]$$

Normalization constant *C* depends on lattice type and space dimension (lattice geometry)

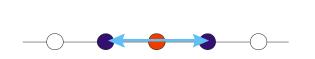
Lattice diffusion – dimension of the space

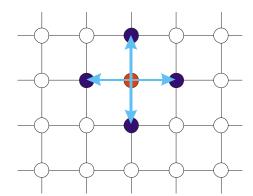
• Simple cubic lattice

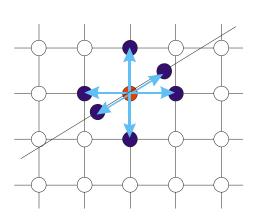
•
$$d = 1$$











•
$$C = 2$$

•
$$C = 4$$

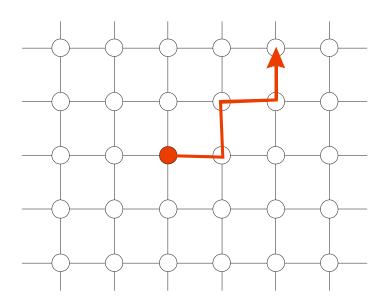
•
$$C = 6$$

• Probability of jump in specific direction $P(i \rightarrow f)$

$$P(i \rightarrow f) = \frac{1}{C} exp \left[-\frac{\Delta E_{diff}}{kT} \right]$$

Displacement in time (tracer diffusion)

• Simple cubic lattice



- Translation vector vectorial sum of the consecutive jumps
- Square of the passed distance (distance difference between position at zero & n jump or sum of the n consecutive jumps

$$\left(\Delta r_{diff}(t)\right)^2 = \left[\vec{r}(t) - \vec{r}(0)\right]^2 = \sum_{i=1}^n \vec{a}_i \cdot \sum_{i=1}^n \vec{a}_i$$

Number of jumps

$$n=\frac{t}{\tau_o}=t\,\nu$$

Distance (tracer diffusion)

Square of the diffusion length

$$\left(\Delta r_{diff}(t)\right)^2 = \sum_{i=1}^{n(t)} \vec{a}_i \cdot \sum_{i=1}^{n(t)} \vec{a}_i = \sum_{\substack{i=1,j=1\\i\neq i}}^{n(t)} \vec{a}_i \cdot \vec{a}_j + \sum_{\substack{i=1}\\i\neq i}}^{n(t)} (\vec{a}_i)^2$$

Ensemble average – over many realizations of the jump paths

$$\left\langle \left(\Delta r_{diff}(t) \right)^{2} \right\rangle = \sum_{\substack{i=1,j=1\\i\neq j}}^{n(t)} \left\langle \vec{a}_{i} \cdot \vec{a}_{j} \right\rangle + \sum_{i=1}^{n(t)} \left\langle (\vec{a}_{i})^{2} \right\rangle$$

Markov process – no correlation between consecutive jumps

$$\langle \vec{a}_i \cdot \vec{a}_j \rangle = \mathbf{0}$$

• Square length is constant

$$(\vec{a}_i)^2 = a^2$$

Mean square displacement in time (tracer diffusion)

Square of the diffusion length in time

$$\left\langle \left(\Delta r_{diff}(t)\right)^{2}\right\rangle = n(t)a^{2}P\left(\Delta E_{diff}\right)$$

Number of jumps

$$n(t) = \frac{t}{\tau_o} = t \nu$$

• Probability of jump $P(\Delta E_{diff})$

$$P(\Delta E_{diff}) = exp\left[-\frac{\Delta E_{diff}}{kT}\right]$$

Distance in tracer diffusion

$$\left\langle \left(\Delta r_{diff}(t)\right)^{2}\right\rangle = n(t)a^{2}exp\left[-\frac{\Delta E_{diff}}{kT}\right] = \frac{a^{2}t}{\tau_{o}}exp\left[-\frac{\Delta E_{diff}}{kT}\right]$$

Tracer diffusion

• Tracer diffusion on surface (d = 2)

$$\left\langle \left(\Delta r_{diff}(t)\right)^{2}\right\rangle = 2dDt = 4D_{sur}t$$

Jumps distance

$$\left\langle \left(\Delta r_{diff}(t)\right)^{2}\right\rangle = a^{2}vt \ exp\left[-\frac{\Delta E_{diff}}{kT}\right]$$

• Diffusion in d = 2 systems

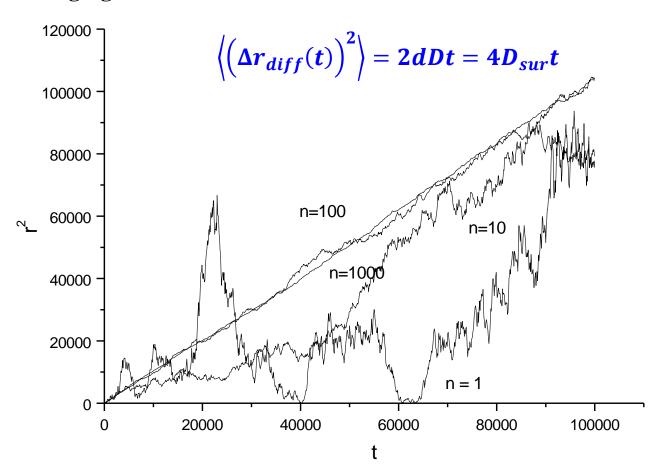
$$D = \frac{a^2 v}{2d} exp \left[-\frac{\Delta E_{diff}}{kT} \right]$$

Diffusion on surface

$$D_{sur} = \frac{a^2 v}{4} exp \left[-\frac{\Delta E_{diff}}{kT} \right]$$

Tracer diffusion in square lattice

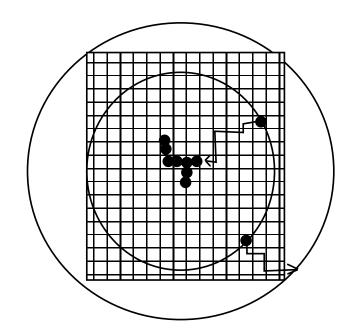
Averaging over defined number of realizations



• Linear dependence is attained in large number of realizations

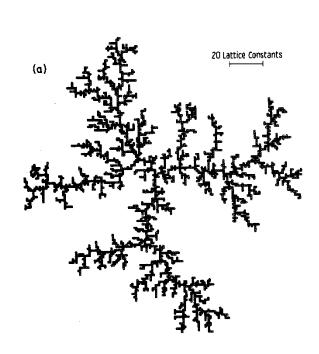
Purely kinetic growth model – diffusion limited aggregation (DLA)

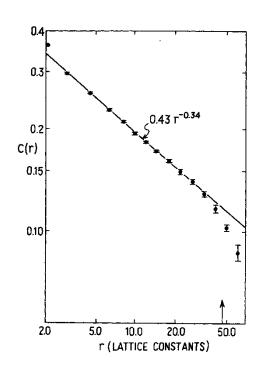
- Molecule start at far distance
- Random jump direction at square lattice
- At contact the molecule is irreversibly included
- Above process is called Diffusion Limited Aggregation

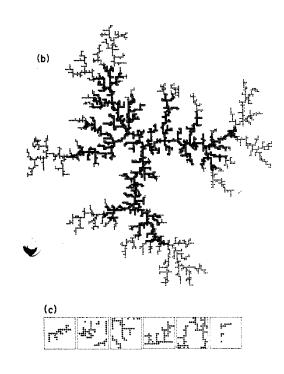


T. A. Witten Jr. & L.M. Sander Phys. Rev. Lett. 47 (1981) 1400

DLA – growth of fractal objects







- Correlation function
- $C(r) = \frac{1}{N} \sum_{r'} \rho(r') \rho(r+r')$

• Geometric dimensionality (Hausdorff)

$$C(r) \sim r^{-0.343 \mp 0.004}$$

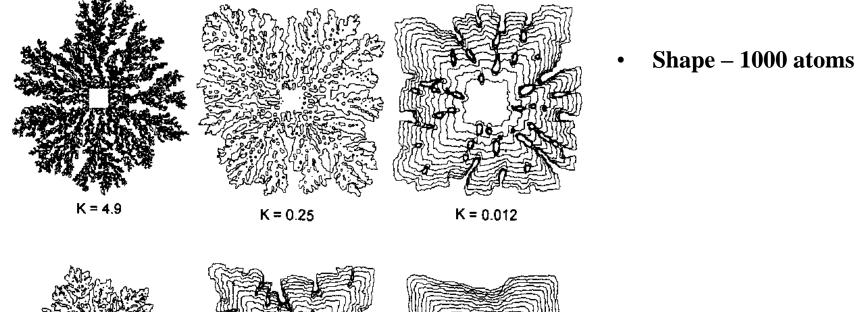
$$D = 1.695 \mp 0.002$$

$$C(r) \sim r^{2-D}$$

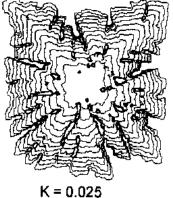
T. A. Witten Jr. & L.M. Sander Phys. Rev. Lett. 47 (1981) 1400

Combined model – transition from fractal to crystal growth

Dense fractal



K = 0.49



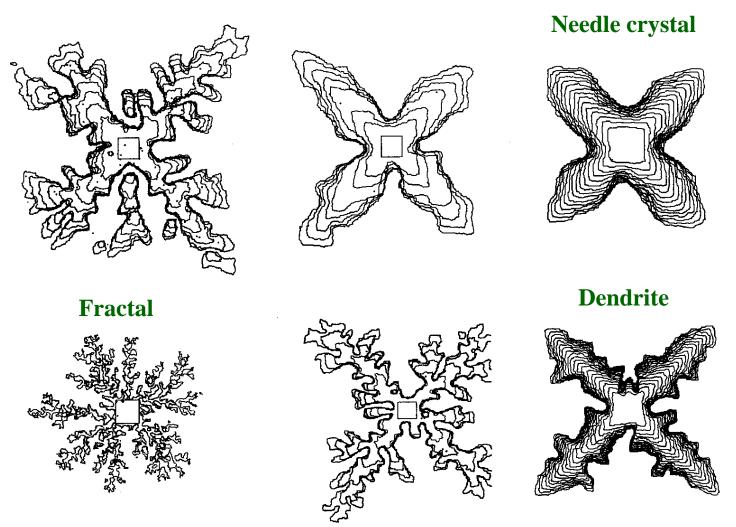
025 K = 0.0012



Crystal

S. Krukowski J.C. Tedenac, J. Cryst. Growth160 (1996) 167

Combined model – transition from fractal to crystal growth



S. Krukowski J.C. Tedenac, J. Cryst. Growth 203 (1999) 269

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